

# Strain-Induced Second Harmonic Generation Enhancement in Monolayer MoS<sub>2</sub> Flakes Edge

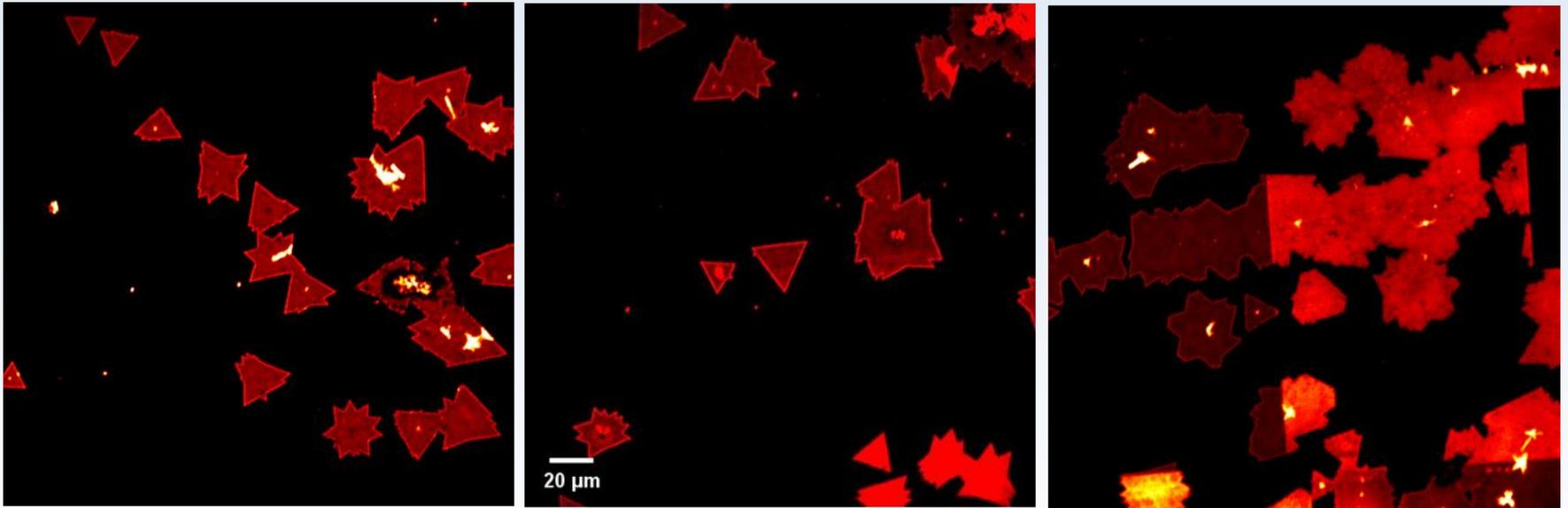
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# Motivation

Second harmonic generation (SHG) intensity mapping using laser scanning confocal microscope (LSCM)

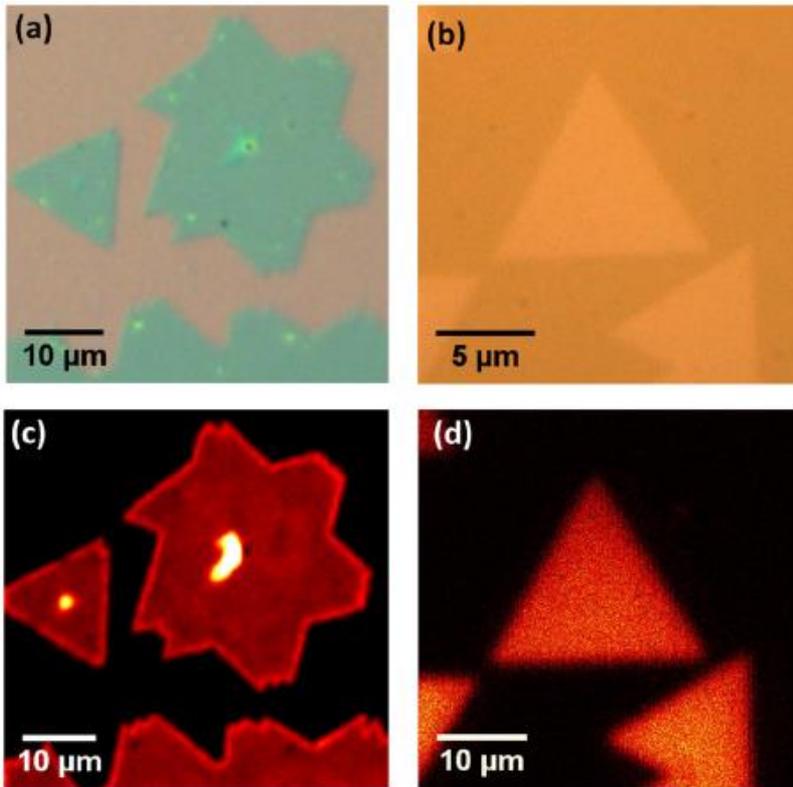


Sample: Monolayer MoS<sub>2</sub> grown on silicon substrate using CVD method

# Abstract

- This edge enhanced SHG may be attributed to the intrinsic strain effect induced during the CVD growth condition, which modulates the nonlinear susceptibility.
- Micro-Raman spectroscopy analysis shows a blue shift in  $A_{1g}$  phonon mode at the flake edge, implies that the flake edge has a compressed and stiffened structure in an out-of-plane direction.
- The AFM analysis of the  $\text{MoS}_2$  flake edge with enhanced SHG, indeed, detected a topographic height change as compared to the center region.

# SHG Intensity Mapping



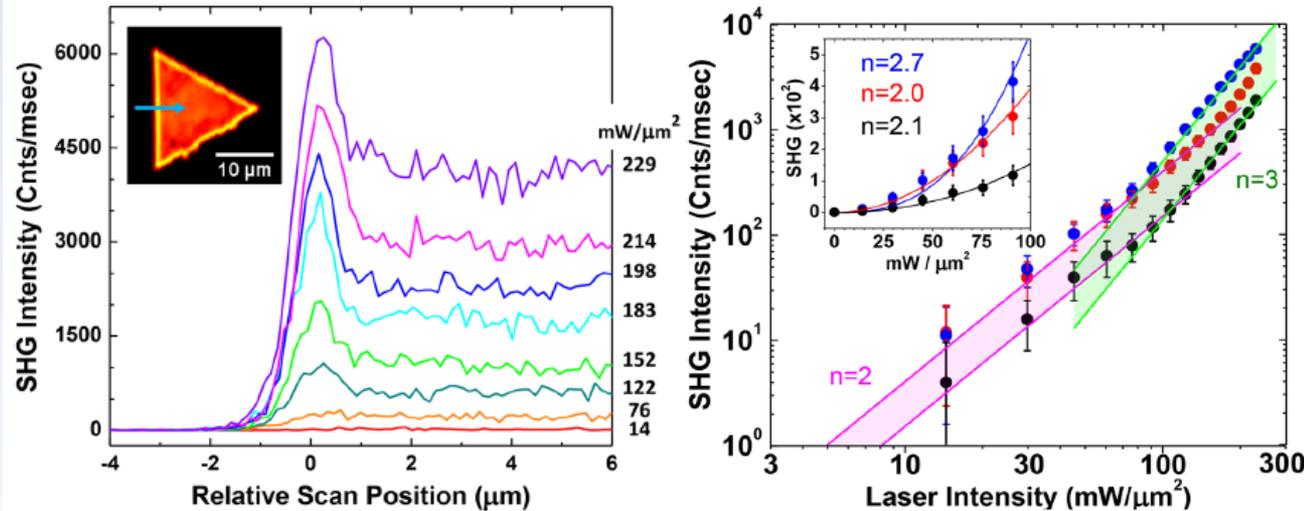
(a)  $\text{MoS}_2@\text{SiO}_2$ : Monolayer  $\text{MoS}_2$  grown on a silicon substrate.

(b)  $\text{MoS}_2@\text{Al}_2\text{O}_3$ : Monolayer  $\text{MoS}_2$  grown on a sapphire substrate.

Selected SHG intensity images of (c)  $\text{MoS}_2@\text{SiO}_2$  and (d)  $\text{MoS}_2@\text{Al}_2\text{O}_3$  acquired using LSCM and 1064 nm femtosecond laser excitation.

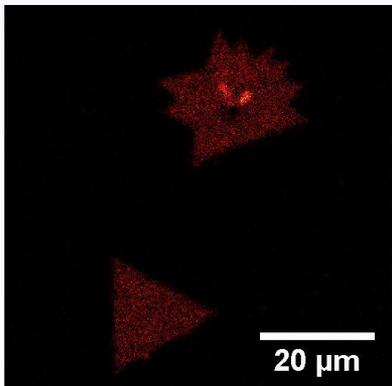
- $\text{MoS}_2 @\text{SiO}_2$  shows an enhanced SHG intensity near the flake edges. However, similar SHG edge enhancement is absent in  $\text{MoS}_2 @\text{Al}_2\text{O}_3$ .

# Power Dependence



The power dependence of SHG intensity of MoS<sub>2</sub>@SiO<sub>2</sub> center region, MoS<sub>2</sub>@SiO<sub>2</sub> edge, and MoS<sub>2</sub>@Al<sub>2</sub>O<sub>3</sub>. The inset figure shows the power dependence in low-intensity range and linear scale.

Red: MoS<sub>2</sub>@SiO<sub>2</sub> flake center.  
 Blue: MoS<sub>2</sub>@SiO<sub>2</sub> flake edge.  
 Black: MoS<sub>2</sub>@Al<sub>2</sub>O<sub>3</sub>.



- The SHG contrast between the edge and the center became significant while increasing the laser intensity.
- A change in the gradient of the power dependence trend line can be observed as the laser intensity increases beyond 100 mW/μm<sup>2</sup>. This may be due to the laser-induced thermal strain on every region of the sample.

# Strain-Dependent SHG Intensity

- The strain dependent SHG intensity at the polarization direction  $\phi$  is

$$I_{\square}^{(2)}(2\omega) \propto (A \cos(3\phi - 3\delta) + B \cos(2\theta + \phi - 3\delta))^2$$

with

$$A = (1 - \nu)(p_1 + p_2)(\varepsilon_{xx} + \varepsilon_{yy}) + 2\chi_0^{(2)}$$

$$B = (1 + \nu)(p_1 - p_2)(\varepsilon_{xx} - \varepsilon_{yy})$$

$\nu$ , Poisson's ratio

$\varepsilon_{xx}$  &  $\varepsilon_{yy}$ , principal stresses

$p_1$  &  $p_2$ , free parameters in photoelastic tensor

$\chi_0^{(2)}$ , nonlinear susceptibility

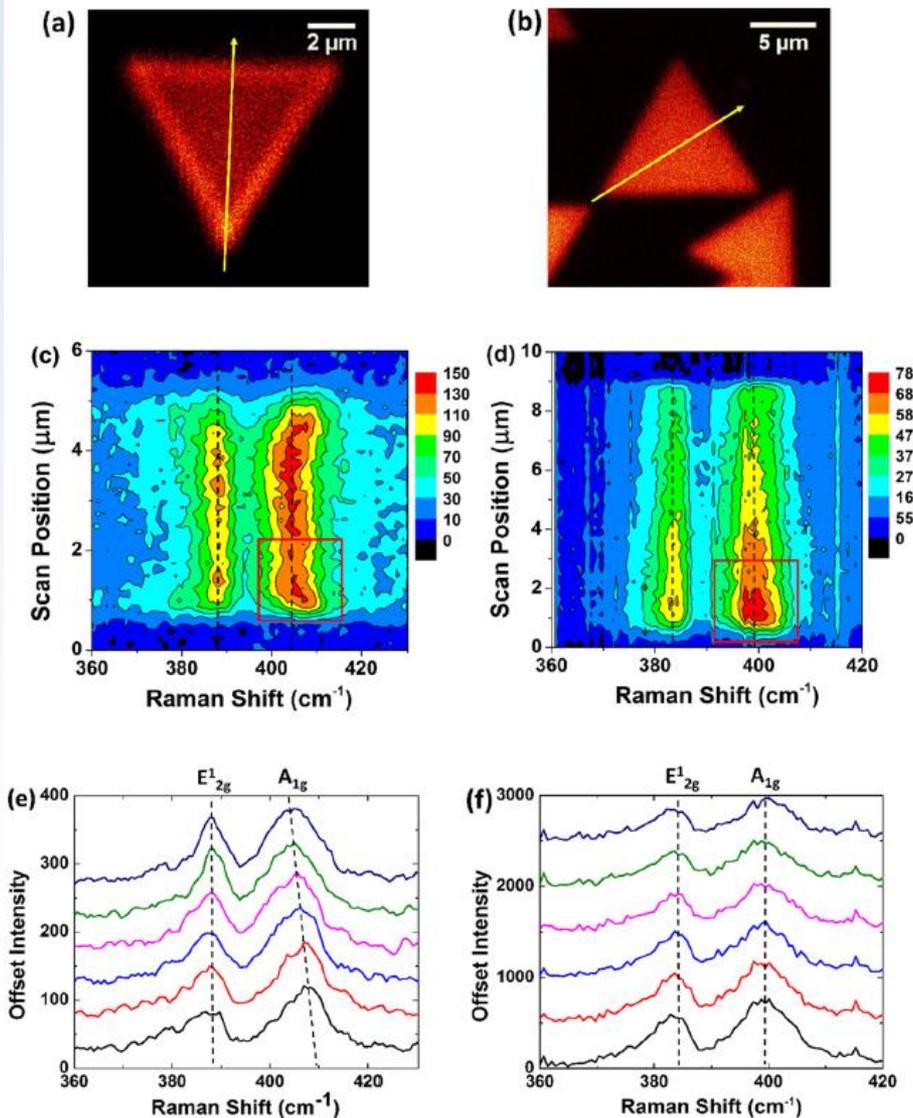
- The SHG variation with strain is related to the sum and difference of the photoelastic tensor elements  $p_1$  and  $p_2$ .
- If the flake edge region is strained anisotropically, the asymmetry in principal strains will result in a nonzero term  $B$ , which further increases the SHG intensity.

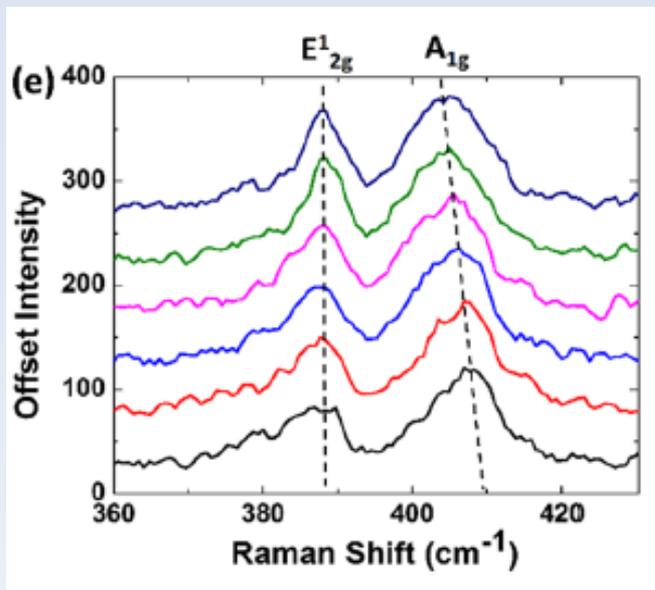
# Micro-Raman Mapping

SHG intensity mapping of (a)  $\text{MoS}_2@ \text{SiO}_2$  and (b)  $\text{MoS}_2@ \text{Al}_2\text{O}_3$ . The yellow arrows show the scanning direction of the line scan Raman spectra.

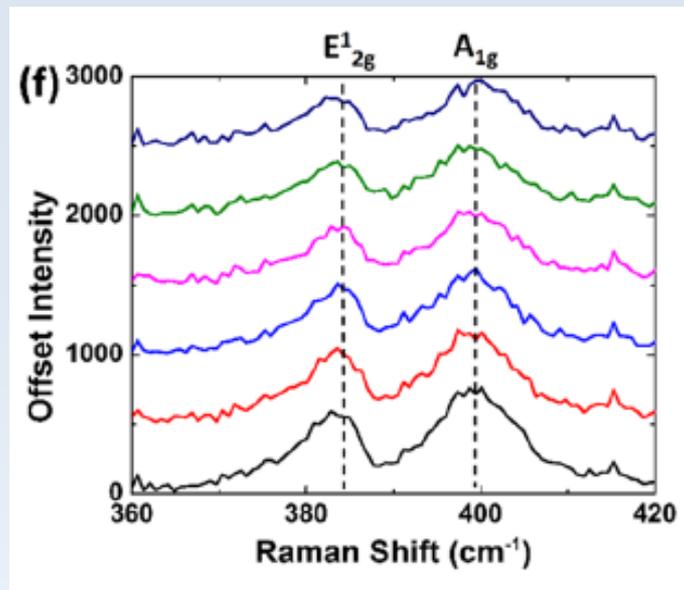
(c) The contour plot of the line scan Raman spectra in  $\text{MoS}_2@ \text{SiO}_2$ . The scan position is following to the scanning direction in (a). (d) The contour plot of the line scans Raman spectra in  $\text{MoS}_2@ \text{Al}_2\text{O}_3$ . The scan position is following to the scanning direction in (b).

(e) The Raman spectra at the edge regions [red box in (c)] of  $\text{MoS}_2@ \text{SiO}_2$ . (f) The Raman spectra at the edge regions [red box in (d)] of  $\text{MoS}_2@ \text{Al}_2\text{O}_3$ .



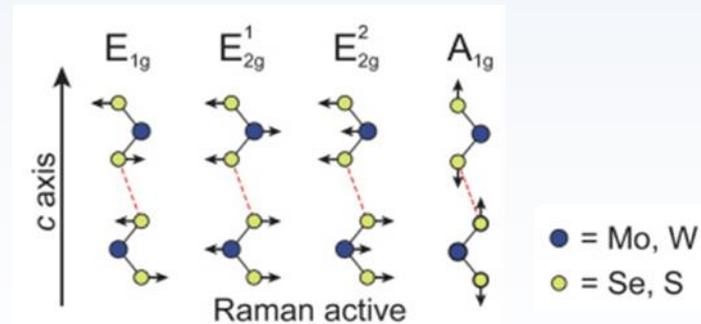


MoS<sub>2</sub>@SiO<sub>2</sub> edge: Blue shift in out-of-plane A<sub>1g</sub> phonon mode, from 404.9 to 407.7 cm<sup>-1</sup> while the in plane E<sup>1</sup><sub>2g</sub> phonon mode remains unchanged at 388 cm<sup>-1</sup>.

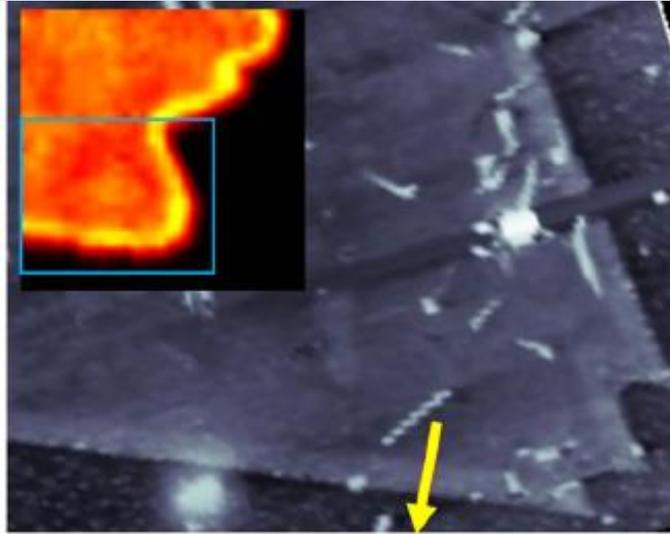


MoS<sub>2</sub>@Al<sub>2</sub>O<sub>3</sub> edge: No shifting in both A<sub>1g</sub> & E<sup>1</sup><sub>2g</sub> phonon mode.

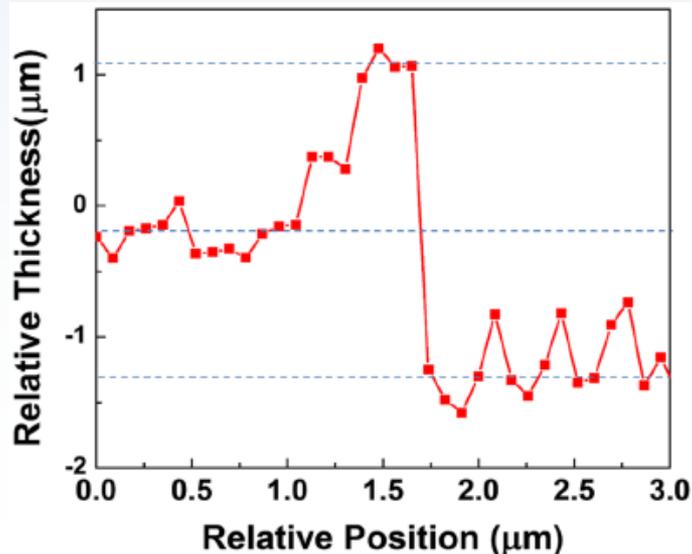
- The A<sub>1g</sub> phonon mode at the flake center shows that it is in unstrained condition, indicates that the structure is compressed and stiffened in the out-of-plane direction at the flake edge.
- The SHG enhancement at the flake edge is related to this compressive strain. There is no SHG enhancement observed in sample MoS<sub>2</sub>@Al<sub>2</sub>O<sub>3</sub> because the strain is rather uniform throughout the sample.



# AFM Mapping



The AFM image of the MoS<sub>2</sub>@SiO<sub>2</sub>. The inset diagram shows the SHG image and the area of interest used in AFM analysis. The width of the white band top is  $\sim 0.34 \mu\text{m}$ .



The topography profile of the flake edge (yellow line). The flake edge has a thickness of  $\sim 1.25 \text{ nm}$  larger than that in flake center region.

- The flake edge is probably a highly strained monolayer as compared to the flake center region.

# Conclusion

- The AFM and Raman spectroscopy results show that the thickness and the phonon modes in the out-of-plane direction at the flake edge in those monolayer MoS<sub>2</sub> flakes are different from the flake center region.
- The intrinsic strain induced during the growth process could be the reason of the edge-enhanced SHG observed in MoS<sub>2</sub>@SiO<sub>2</sub>. The laser-induced thermal effect could further alter the power dependence behavior of the SHG signal from every sample.
- These results bring out the awareness of the residual strain effect on 2D TMD device performance and the adequate characterization of the strained structure by SHG intensity mapping.

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Thank you